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## **Mg<sub>2</sub>Si film on Si(111) prepared by Ultra-Fast Mg reactive deposition: crystal structure and thermoelectric properties**

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**Abstract.** The Mg<sub>2</sub>Si film (~ 800 nm thick) was grown by pulsed reactive deposition of Mg on Si(111) at 340 °C in UHV. Structural investigations by XRD, SEM and cross-sectional x-HR-TEM demonstrate high crystal quality and 100% texture of the film. Thermoelectric properties of the Mg<sub>2</sub>Si film are characterized within 290–470 K. The film conductivity changes from p-type below 309 K to n-type at higher temperatures. The power factor is 0.27 mW/m×K<sup>2</sup> at 470 K. The p-type conductivity can be associated with presence of oxygen or/and vacancies (V<sub>Mg</sub>, V<sub>Mg2Si</sub>).

**Keywords:** magnesium silicide, silicon, films, epitaxy, reactive epitaxy, pulsed deposition, crystal structure, microscopy, transport properties, Seebeck coefficient, power factor

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Материалы конференции

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## **Пленка Mg<sub>2</sub>Si на Si(111), полученная методом сверхбыстрого реактивного осаждения Mg: структура и термоэлектрические свойства**

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**Аннотация.** Пленка  $Mg_2Si$  толщиной  $\sim 800$  нм была выращена методом импульсного реактивного осаждения Mg на Si(111) при  $340^\circ C$  в условиях сверхвысокого вакуума. Структурные исследования с помощью рентгеновской дифракции, сканирующей электронной микроскопии и высокоразрешающей просвечивающей микроскопии поперечного сечения образца демонстрируют высокое кристаллическое качество и 100% текстуру пленки. Термоэлектрические свойства пленки  $Mg_2Si$  были исследованы в диапазоне температур от 290 К до 470 К. Тип проводимости пленки меняется с p-типа проводимости при температурах ниже 309 К на n-тип при более высоких температурах. Фактор мощности составляет  $0,27$  мВт/м $\times$ К $^2$  при 470 К. Проводимость p-типа может быть связана с наличием кислорода, и/или вакансиями Mg и тройными вакансиями  $Mg_2Si$ .

**Ключевые слова:** силицид магния, кремний, пленки, эпитаксия, реактивная эпитаксия, импульсное осаждение, кристаллическая структура, микроскопия, транспортные свойства, коэффициент Зеебека, фактор мощности

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## Introduction

The narrow band ( $E_g \sim 0.6\text{--}0.8$  eV) semiconductor  $Mg_2Si$  is a promising material for SWIR detectors and solar cells [1] and thermoelectric (TE) energy conversion [2]. Usually,  $Mg_2Si$  is an n-type semiconductor, almost irrespective to a synthesis method. The p-type  $Mg_2Si$  is highly required for preparing p- $Mg_2Si$ /n-Si heterojunction SWIR sensors and p-type legs for thermoelectric (TE) double leg generators based on n- and p- $Mg_2Si$ .  $Mg_2Si$  of the p-type has been obtained by overcompensation with impurities, e.g. Ag, but this resulted in a significant decrease in carrier mobility [3]. As well, the p-type conductivity can be caused by vacancies ( $VMg$ ,  $VMg_2Si$ ) or oxygen interstitials which represent acceptors [4–6]. Synthesis of high crystal quality  $Mg_2Si$  is a difficult task. The  $Mg_2Si$  films grown on Si substrates at  $\approx 200^\circ C$  [7, 8] have rather poor crystal quality, irrespective of the growth technique. For typical deposition rates  $\sim 10^{-1} - 10$  nm/min and substrate temperatures (T) above  $\approx 300^\circ C$ , Mg re-evaporates without formation of  $Mg_2Si$ . We showed [10] that this problem can be avoided by radical increase of the deposition rate. In our works [9, 10]  $\sim 10\text{--}100$  nm thick perfect  $Mg_2Si$  films were grown on Si(111) and Si(100) substrates at  $360\text{--}480^\circ C$  by pulsed UHV deposition of Mg at the rates of  $10^2\text{--}10^4$  nm/sec. Since the temperature of this exothermic process is in large extent determined by the reaction heat release ( $\approx 0.8$  eV per a formula unit) we concluded that the initial temperature of  $\approx 340^\circ C$  can be optimal for  $Mg_2Si$  film growth. In this work we present the growth and TE properties of a  $\sim 800$  nm thick crystalline  $Mg_2Si$  film prepared on Si(111) by ultra-fast Mg deposition ( $\approx 10^4$  nm/s) at the substrate temperature  $340^\circ C$ .

### Materials and Methods

The experiment was carried out in the Ultra High Vacuum (UHV) chamber Varian with the base vacuum of  $\sim 2 \times 10^{-9}$  Torr. The substrate, Si(111)  $\rho > 1000$  Ohm $\times$ cm was degassed in UHV by DC heating at  $\sim 650$  °C for 6 hours and cleaned by multiple high temperature ( $T$ ) flashes at  $T \sim 1120$  °C just before the growth experiment. The magnesium source was well degassed during several hours at high  $T$  in UHV as well. The pyrometer PhotriX was used for  $T$  measuring. The calibration and pulse deposition procedures are described in our previous works [5, 6]. The Mg portion ( $\approx 650$  nm) was deposited onto the substrate during  $\approx 200$  msec. Some small part of the deposited dose re-evaporated from the hot substrate (340 °C) but most of Mg atoms reacted with Si and finally a  $\sim 800$  nm thick  $Mg_2Si$  film was synthesized. The XRD spectrum was obtained with a RIGAKU SmartLab diffractometer using a 9 kW rotating anode,  $CuK\alpha$  radiation. The diffraction patterns were recorded within the  $2\theta$  range of  $22^\circ - 61^\circ$  with the  $2\theta$  step of  $0.01^\circ$ . The thickness of the grown  $Mg_2Si$  film was studied with the ThermoFisher Scios 2 DualBeam scanning electron microscope (SEM). For preparing a hole with a smooth flat surface a focused  $Ga^+$  ion beam (FIB) was used. The Cross-sectional High-Resolution Transmission Electron Microscopy (x-HRTEM) image was obtained with an electron microscope JEM-2100 Plus JEOL operating at 200 kV. The thermoelectrical measurements (Seebeck coefficient and electrical conductivity) were carried out in He atmosphere in the temperature range 290–470 K in a Cryotel laboratory setup (MISIS).

### Results and Discussion

Fig. 1 demonstrates Raman spectroscopy data for the grown  $Mg_2Si$  film. The strong  $F_{2g}$  Raman peak at  $257\text{ cm}^{-1}$  and absence of the peak at  $520\text{ cm}^{-1}$  attributed to crystalline Si indicate a rather thick  $Mg_2Si$  film. Some other  $Mg_2Si$  Raman modes are also clearly seen in Fig 1. Raman spectra demonstrate that the positions of all the peaks are slightly shifted to low wavenumbers as compared to the literature data [11, 12] (see Table). We suggest that this indicates some compressive stresses inside the  $Mg_2Si$  film. The stress could occur during the very short heating-cooling cycle due to the large difference in the thermal expansion factors; as well it could be caused by the lattice mismatch.

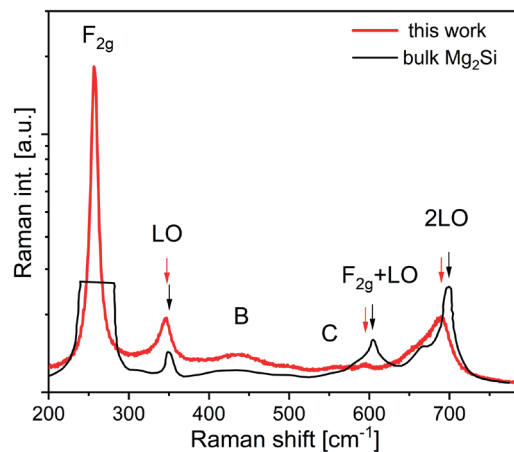


Fig. 1. Raman scattering spectrum for  $Mg_2Si$  film on Si(111) substrate (red curve) and reference [11] data for bulk  $Mg_2Si$  (black curve)

Table

The positions of the peaks of the Raman spectrum

Mode	Raman shift[ $cm^{-1}$ ]	
	this work	Bulk $Mg_2Si$ [11]
$F_{2g}$	257	258.5 [12]
LO	346	360
B	433	433
$F_{2g} + LO$	568	604
2LO	689	699

Fig. 2, *a* shows the XRD data of the  $\text{Mg}_2\text{Si}/\text{Si}(111)$  sample. The spectrum demonstrates only peaks from silicon substrate (111) and  $\text{Mg}_2\text{Si}$  features corresponding to (111) and (222) planes at  $2\theta$  angles  $24.28^\circ$ ,  $49.79^\circ$  respectively. No other magnesium silicide phases ( $\text{Mg}_5\text{Si}_6$ ,  $\text{Mg}_9\text{Si}_5$ ,  $\text{Mg}_3\text{Si}$ ) are observed. The  $\text{Mg}_2\text{Si}$  lattice parameter ( $6.346 \text{ \AA}$ ) obtained from our data is  $\sim 0.08\%$  smaller than that accepted in [13] ( $6.351 \text{ \AA}$ ). This supports the above suggestion on some compression stress in the film material. This result is in correspondence with the Raman investigations data discussed above.

By SEM data, the film has the thickness  $\approx 800 \text{ nm}$  (see insertion in Fig. 2, *a*). The cross-sectional HRTEM image is presented in Fig. 2, *b*. The interface  $\text{Mg}_2\text{Si}/\text{Si}$  is sharp. The typical columnar structure [5] is clearly seen. The column lateral size varies within  $100\text{--}170 \text{ nm}$ . The upper surface of the film is naturally oxidized.

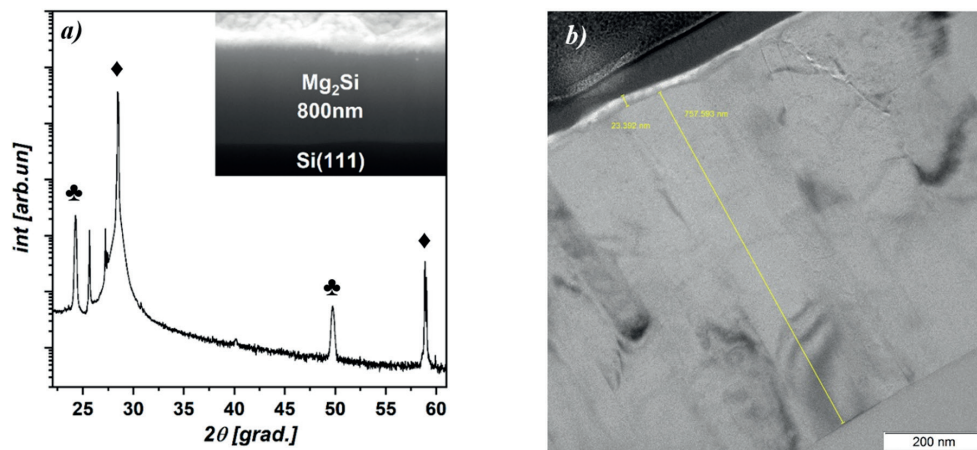


Fig. 2. The XRD spectrum of the  $\text{Mg}_2\text{Si}/\text{Si}(111)$  sample and the SEM cross-section profile in insertion,  $\approx 800 \text{ nm}$   $\text{Mg}_2\text{Si}$  film demonstrates high crystal quality and perfect texture (111).  $\text{Mg}_2\text{Si}$  and  $\text{Si}(111)$  peaks are indicated with ♣ and ♦ marks, respectively (*a*). The cross-section HRTEM image, the  $760 \text{ nm}$  thickness and columnar structure are clearly seen (*b*)

The temperature dependencies of the thermoelectric properties of the  $\text{Mg}_2\text{Si}$  film in the range of  $290\text{--}470 \text{ K}$  are shown in Fig. 3. The Seebeck coefficient is  $+478 \mu\text{V}\times\text{K}^{-1}$  at  $290 \text{ K}$  (see Fig. 3, *a*), it decreases with  $T$  and changes the sign at  $309 \text{ K}$ . In the range of  $350\text{--}470 \text{ K}$  its dependence on  $T$  is linear. At  $470 \text{ K}$  it gets the value of  $-605 \mu\text{V}\times\text{K}^{-1}$ . The Seebeck coefficient sign change corresponds to the compensation between p- and n-type carriers. At  $309 \text{ K}$  the intrinsic n-type conductivity of the  $\text{Mg}_2\text{Si}$  film is activated and the electron contribution becomes larger than the hole contribution. The nature of the p-type conductivity in  $\text{Mg}_2\text{Si}$  is under investigation now. Theoretical calculations show that Mg vacancies,  $\text{Mg}_2\text{Si}$  triple vacancies and oxygen interstitials represent acceptors in  $\text{Mg}_2\text{Si}$  [8–10].

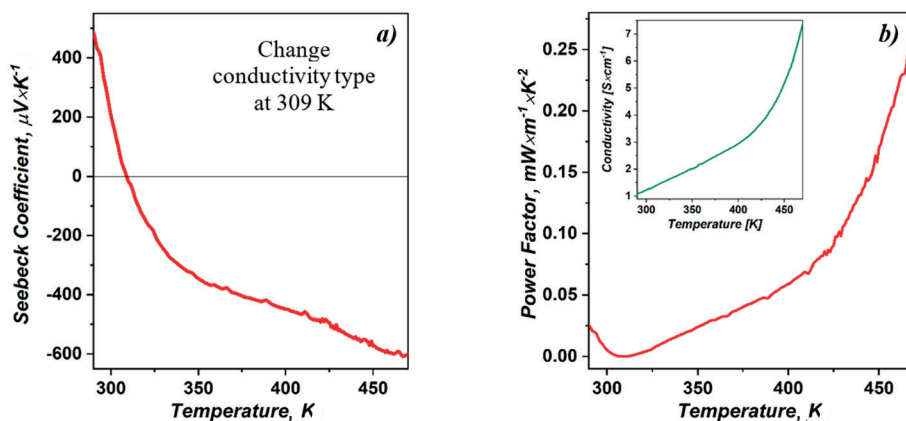


Fig. 3. Temperature dependence of Seebeck coefficient (*a*), Power Factor (*b*) and conductivity (insertion on *b*) for  $\text{Mg}_2\text{Si}$  film on Si substrate

The conductivity of the  $\text{Mg}_2\text{Si}$  film increases with temperature (see insertion in Fig. 3, *b*) and reaches  $7.4 \text{ S}\cdot\text{cm}^{-1}$  at 470 K. As seen in Fig. 3, *b*, the Power Factor (PF) maximum is  $0.27 \text{ mW}/\text{m}\cdot\text{K}^2$  at 470 K. The PF is rather small at 290 K ( $25 \mu\text{W}/\text{m}\cdot\text{K}^2$ ), decreases almost to zero at 309 K and then increases with  $T$ . This behavior and the PF value around 309 K are due to carrier compensation. Further investigations of the thermal conductivity and effect of impurities on thermoelectric properties are planned for our future researches.

### Conclusions

High crystal quality  $\sim 800 \text{ nm}$  thick  $\text{Mg}_2\text{Si}$  film with  $\approx 100\%$  texture (111) was formed by pulsed Mg deposition onto Si(111) at  $340^\circ\text{C}$ . Some compressive stress of  $\approx 0.08\%$  is present in the film material. The film conductivity is of the p-type at  $T < 309 \text{ K}$  and n-type at  $T > 309 \text{ C}$ . The p-type could be attributed to Mg vacancies,  $\text{M}_2\text{Si}$  triple vacancies and oxygen interstitials. The power factor is  $0.27 \text{ mW}/\text{m}\cdot\text{K}^2$  at 470 K.

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